

# High Surface Area Iron Oxide Microspheres via Ultrasonic Spray **Pyrolysis of Ferritin Core Analogues**

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# Supporting Information

The controlled synthesis and fabrication of nano- and microstructured materials has received much attention over the past decade owing to the unique chemical, electrical, magnetic, and optical properties of nanoscaled materials when compared to the bulk. Due to its low price and abundance, low toxicity, biodegradability, and high chemical stability, iron oxide is an especially promising material which has been studied for use in applications such as photocatalysis, <sup>1-6</sup> heavy metal removal, <sup>7,8</sup> sensing, <sup>9-11</sup> and energy storage. <sup>4,10-14</sup>

Previously, high surface area, nanostructured iron oxides have been synthesized by using a template, such as polystyrene microspheres<sup>15</sup> or copper nanowires, <sup>16</sup> on which iron oxide or an iron oxide precursor is grown. These templates can be expensive and are typically destroyed upon removal. Forced hydrolysis of an iron precursor in the presence of surfactants can also lead to nano- or microspheres of iron oxide with varying surface areas. 17,18 Iron oxide nanodiscs have been made using a controlled etching with oxalic acid. 19 The most common method for producing iron oxide microspheres has been solvothermal. <sup>6,7,9,20–26</sup> Some of these procedures require harsh chemicals or expensive surfactants, while other try to minimize the environmental impact of production. All, however, are batch processes that take many days in a high pressure vessel and generally require further workup; in addition, all of these methods are expensive (often prohibitively) and are difficult to scale-up.

Spray pyrolysis and similar aerosol techniques are wellknown as scalable synthetic methodologies for the preparation of metal oxide materials from relatively inexpensive precursors. 27-32 Porous particles with relatively high surface areas have been obtained with spray pyrolysis for carbon, <sup>33–40</sup> silica, <sup>41,42</sup> titania, <sup>43</sup> alumina, <sup>44</sup> Bi<sub>2</sub>WO<sub>6</sub>, <sup>45</sup> Mn<sub>3</sub>O<sub>4</sub>, <sup>46</sup> ZnS, <sup>47</sup> and MoS<sub>2</sub>. <sup>48</sup> Iron oxides, however, have not been previously prepared in high surface area form through spray pyrolysis; only films 49,50 and solid particles 28,51-54 have been reported.

In this work, we show the use of ultrasonic spray pyrolysis  $(USP)^{31,32}$  to create very high surface area, porous, iron oxide microspheres. Through the use of colloidal oxy/hydroxyiron(III) polymers in the precursor solution (ferritin core analogues, so-called Spiro-Saltman balls<sup>55,56</sup>), porous iron oxide spheres are created that are more crystalline than those made from simpler iron salt precursors. Due to the nature of USP, the size of the particles can be controlled without affecting the properties of the material; the morphology and surface area can be easily tuned by changing the salts used to prepare the precursor. This method is an inexpensive and facile way to synthesize high surface area, porous, iron oxide microspheres

from simple and inexpensive precursors using the continuous, scalable process.

First, a solution of ferritin core analogues was prepared by reacting aqueous solutions of Fe(NO<sub>3</sub>)<sub>3</sub>·(H<sub>2</sub>O)<sub>9</sub> (98%, ACS from Sigma-Aldrich) with Na<sub>2</sub>CO<sub>3</sub> or NaHCO<sub>3</sub> solutions. The mixture was stirred and sonicated in a cleaning bath until no more gas evolved (<5 min); this was easily monitored visually as the initially yellow Fe<sup>3+</sup> solution turned opaque and brown as CO<sub>2</sub> gas evolved, eventually becoming a translucent dark red. DI water was added to make a 50 mL solution.

The precursor solutions were then nebulized using a madein-house nebulizer (1.65 MHz,  $\sim 10 \text{ W/cm}^2$ ) connected to a quartz flow tube inside a tube furnace, as described previously. 37,38 Compressed air flowing at 1 SLPM was used as a carrier gas to carry the precursor droplets through a furnace tube at 500 °C (residence time ~10 s). The final product was collected in a series of water bubblers and washed 3× with water to remove any remaining salt porogen. During the heating of the droplets, nitrate ions decompose quickly, forming a NaOH salt template in situ and yielding NO<sub>x</sub> gases which act as porogens.<sup>35</sup> The NaOH template is removed in the water bubblers and during subsequent washings leaving highly porous iron oxide spheres.

From a precursor solution containing 0.2 M Fe(NO<sub>3</sub>)<sub>3</sub> and  $0.2 \text{ M Na}_{2}\text{CO}_{3}$ , spheres  $560 \pm 180 \text{ nm}$  in diameter are formed (Figure 1a and Supporting Information Figure S1a). TEM micrographs suggest the spheres are porous throughout (Figure 1b), and Brunauer-Emmett-Teller (BET) measurements show a surface area of 301 m<sup>2</sup>/g and average pore radius (as determined using the Barrett-Joyner-Halenda method) of 2.1 nm. This surface area is as high as any yet recorded for an iron

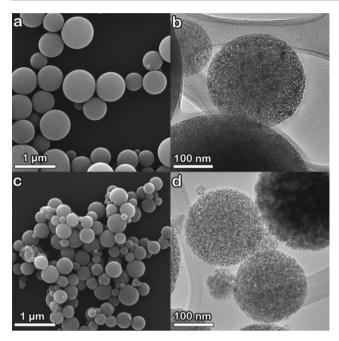
Powder X-ray diffractometry (XRD) confirms the presence of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> with an average crystallite size of 7.1 nm (Figure 2). Elemental analysis (EA) confirms the removal of the Na (<1 wt %). As a control, particles were made from a precursor composed of only Fe(NO<sub>3</sub>)<sub>3</sub> and different amounts of the porogen NaNO<sub>3</sub> (i.e., no Spiro-Saltman balls). Compared to the Spiro-Saltman-based spheres, the control spheres (spheres formed using Fe(NO<sub>3</sub>)<sub>3</sub> and a neutral salt porogen NaNO<sub>3</sub>) had similar surface areas but decreased crystallinity (Figure 2). Conversely, by using a strong base (NaOH) in place of the weakly basic Na<sub>2</sub>CO<sub>3</sub> or NaHCO<sub>3</sub> to make the precursor

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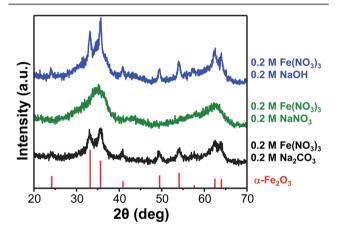


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**Figure 1.** SEM (left) and TEM (right) micrographs of microspheres prepared from (a, b) 0.2 M Fe(NO<sub>3</sub>)<sub>3</sub> and 0.2 M Na<sub>2</sub>CO<sub>3</sub> and (c, d) 0.02 M Fe(NO<sub>3</sub>)<sub>3</sub> and 0.02 M Na<sub>2</sub>CO<sub>3</sub>.



**Figure 2.** XRD diffractograms of the porous microspheres made from (black) Spiro-Saltman ferritin core analogues and two controls: (blue) a neutral salt and (green) a stronger base. Hematite peaks (red) are included as a reference (PDF no. 04-003-2900).

solution, larger iron oxide particles are formed resulting in relatively high crystallinity (Figure 2) but a lower surface area  $(206 \text{ m}^2/\text{g})$ . The use of ferritin core analogues over simple iron

salts as precursors produces products with greater crystallinity and similarly high surface areas.

The average size of the spheres is tunable by changing the concentration of the precursors. For example, spheres 280  $\pm$  110 nm in diameter were synthesized by using 0.02 M  $\rm Fe(NO_3)_3$  and 0.02 M  $\rm Na_2CO_3$  (i.e., concentrations an order of magnitude smaller than that used for the original spheres) (Figure 1c and Supporting Information Figure S1b). These smaller spheres have similar properties to those of the larger spheres. Again, TEM suggests porosity throughout the entire particle (Figure 1d), and BET measurements confirm a similar surface area (294  $\rm m^2/g)$  and average pore radius (1.7 nm). EA confirms the removal of Na.

Other Fe3+ salts can also be used to prepare the Spiro-Saltman precursor. A precursor obtained by reacting FeCl<sub>3</sub> and Na<sub>2</sub>CO<sub>3</sub> results in hollow, porous spheres (Figures 3e, Supporting Information Figures S2d and S3c). Here, the NaCl byproduct acts as an in situ salt template; however, unlike NaNO3, no decomposition occurs, and the salt template is a solid instead of a liquid as is the case for the NaOH remaining after NaNO $_3$  decompostion (MP $_{NaCl}$  = 801 °C and MP $_{NaOH}$  = 318 °C).<sup>37</sup> The lack of the  $NO_x$  gas porogen can account for the lower surface area (97 m<sup>2</sup>/g) of the product. Particles with intermediate morphologies and surface areas are created by simply mixing different molar ratios of Fe(NO<sub>3</sub>)<sub>3</sub> and FeCl<sub>3</sub> (Figure 3b-d, Supporting Information Figures S2a-c and S3ab). EA for all of the microspheres show negligible Na, indicating that the salt templates were completely removed with washing. Although the TEM for the 3:1 and 1:1 Fe(NO<sub>3</sub>)<sub>3</sub>/FeCl<sub>3</sub> samples appear similar to the Fe(NO<sub>3</sub>)<sub>3</sub> only sample, the differences in surface areas indicate a decrease in porosity as more FeCl<sub>3</sub> is added. It is hypothesized that both the decrease in release of NO<sub>x</sub> porogen and the solid salt template (NaCl) present with FeCl<sub>3</sub> rich precursors leads to larger voids.

Hematite has been suggested for use as a lithium-ion battery anode given its high capacity, environmental benignity, abundance, and low cost. Highly porous iron oxide materials increase the practical capacity of anodes by increasing the available material for lithiation. As a proof of concept, we tested our iron oxide microspheres as Li-ion battery anodes. Figure 4 shows the resulting charge capacities, discharge capacities, and Coulombic efficiencies for 10 cycles of a half-cell of the iron oxide-based working electrode versus a Li counter electrode for the 0.2 M Fe(NO<sub>3</sub>)<sub>3</sub>/0.2 M Na<sub>2</sub>CO<sub>3</sub> spheres. As with many iron oxide anode materials, there is a very large initial charge capacity usually attributed to surface electrolyte interface layer formation. The charge and discharge capacities diminish over only a few cycles and reach an ultimate efficiency

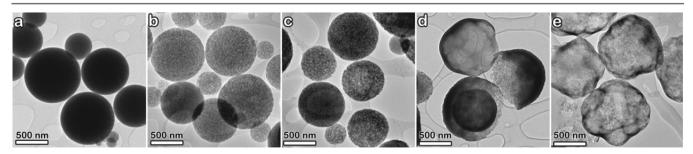
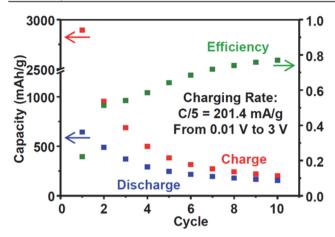


Figure 3. TEM micrographs of microspheres made from different molar ratios of Fe(NO<sub>3</sub>)<sub>3</sub>/FeCl<sub>3</sub>: (a) Fe(NO<sub>3</sub>)<sub>3</sub> only, (b) 3:1, (c) 1:1, (d) 1:3, and (e) FeCl<sub>3</sub> only. Scale bars are 500 nm. BET surface areas are (a) 301 m<sup>2</sup>/g, (b) 226 m<sup>2</sup>/g, (c) 176 m<sup>2</sup>/g, (d) 132 m<sup>2</sup>/g, and (e) 97 m<sup>2</sup>/g.

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**Figure 4.** Charge capacity (red), discharge capacity (blue), and efficiency (green) over 10 cycles for a half-cell made with porous iron oxide from a 0.2 M  $Fe(NO_3)_3/0.2$  M  $Na_2CO_3$  precursor solution. Working electrode is 50% iron oxide, 40% conductive carbon, and 10% polyvinylidene fluoride binder by weight on a copper current collector, counter and reference electrodes are Li, and electrolyte is 1 M LiPF<sub>6</sub> in 1:1 v/v ethylene carbonate/dimethyl carbonate.

of just under 80% after 10 cycles. These results show that these materials may prove useful as anodes in lithium-ion batteries but that further development is necessary to become competitive with the current state-of-the-art. 4,58,59

In conclusion, we have shown that highly porous iron oxide spheres of different morphologies, sizes, and crystallinities can be produced from simple and inexpensive precursors using the continuous, scalable process of ultrasonic spray pyrolysis. The nebulized precursor solution is based on colloidal iron oxy/hydroxy polymers (ferritin core analogues, so-called Spiro-Saltman balls) which allows for an increased crystallinity while maintaining a high surface area.

# ASSOCIATED CONTENT

#### Supporting Information

Size distribution for Figure 1a,c; higher magnification TEM of Figure 3b–e; SEM of Figure 3b,c,e. The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.chemmater.5b00766.

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#### Notes

The authors declare no competing financial interest.

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